Solidification Behavior of Water-Cooled and Subzero-Chilled Cast Iron on Mechanical Properties

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Hypoeutectic cast iron specimens cast using chills that are water-cooled and liquid-nitrogen-cooled (subzero chilling) were compared with specimens of the same chemical composition, which were sand-cast without any chill. The solidification behavior, number of eutectic cells, grain size, and effects of these on the mechanical properties such as strength and fracture toughness were recorded and analyzed in this paper. It is revealed from the above investigation that subzero and water-cooled chilled cast irons exhibit

severe under cooling as compared to that of normal sand-cast iron. Thus, it is concluded from the investigation that nucleation conditions are completely altered during solidification, which is considered to be responsible for the variation in eutectic cell size, grain size, microstructure, and hence mechanical properties of the cast iron.

Chilled cast iron belongs to a group of metals possessing
thigh strength, hardness toughness, and wear resistance. For a
metallurgist, there is sufficient information available on the
metallurgist, there is sufficient info relationship between these parameters in the case of water-
cooled and liquid-nitrogen-cooled chilled cast irons. The reason
size, and resulting microstructures with various cooling behind the selection of this series of chilled cast iron for the rates;
present investigation is that a wide range of strength and fracture present investigation is that a wide range of strength and fracture
to analyze the data in light of the solidification process; and
to analyze the data in light of the solidification process; and
to correlate ultimate tens

1.2 Effect of Cooling Rate during Solidification

In general, the cooling rate during casting is largely governed **1.3 Past Research** by the design and thermal nature of the casting procedure, Several properties of cast iron are of interest to operating one significant factor being the mold material. Metal molds foundrymen, including UTS, fracture toughness, and hardness. generally offer higher chilling action on the solidifying mass These properties are influenced mainly by chemical composidue to their heat diffusivity. Therefore, the influence of higher tion and microstructural features, including graphite type, cooling rate is normally responsible for the superior properties graphite content, and cell count. As most foundry metallurgists of chilled castings. The influence of very high cooling rates in know, the composition of the cast iron determines the quantity producing fine structures offers the possibility of future devel-
and character of the graphit opment of cast irons possessing high strength and fracture teristics of the metallic matrix for some specified set of cool-
toughness. The undercooling of a melt to a lower temperature ing conditions.^[1] toughness. The undercooling of a melt to a lower temperature

Keywords chill, solidification, subzero, water-cooled increases the number of effective nuclei for solidification relative to the growth rate, the latter being restricted by the rate at which the latent heat of crystallization can be dissipated. **1. Introduction** Conversely, slow cooling favors the growth from a few solidification nuclei and produces coarse grain structures. The refining **1.1 Chilled Cast Iron** effect of enhanced cooling rate applies both to the primary grain size and to the substructure, although in the latter case,

- size, and resulting microstructures with various cooling
-
- microstructure.

and character of the graphite, as well as the metallurgical charac-

Cooling curves of cast iron have been used quite successfully to explain the solidification behavior of gray cast iron (stable) **Joel Hemanth,** Department of Mechanical Engineering, Siddaganga
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Contact e-mail: joelhemanth@usa.net.
Contact e-mail: joelhemanth@usa.net.
Contact e-mail: joe irons in general.^[2,3] Significant undercooling reflects an absence

Table 1 Composition of cast iron tested

Element	C	Si	Mn		Cu	Fe
%Composition 3.42 1.8 0.41 0.04 0.08 1.57						Balance

of nuclei for solidification, which can serve as eutectic cell centers.

In a study of the fracture mechanics of gray cast iron, Baker^[4] observed that both the UTS and fracture toughness decreased linearly with increasing eutectic cell size. Suzuki and Kayama^[5] concluded from their own investigations that graphite nucleation out of residual graphite is by means of eutectic cells and that eutectic cell number is an index to nucleation frequency. DeHoff and Rhines^[6] developed an equation to calculate the number of eutectic cells per unit volume, and this can be deduced using the number of eutectic cells. Fedrikson and $Hillert^[7] studied the graphite precipitation process in an allowed$ cast iron under conditions of controlled solidification at various rates. **Fig. 1** Mold used for casting specimens Nucleation of the eutectic liquid can be a difficult process

since considerable undercooling is often observed before solidification begins. If the stable eutectic is not nucleated, then the iron may undercool and solidify to form iron carbides mixed same type of mold was used to sand-cast specimen C, in which

dritic structure that solidifies hosts of the cells and therefore (at -60° C), respectively. can have a major effect on cell nucleation and growth.[9] Ruff and Wallace^[10] showed that, by increasing the eutectic cell count, the effective span and stress concentration effect of the **2.3 Specimen Testing** graphite was reduced and the UTS was improved. A suggested
classification of cell size has been proposed by Dawson and
Oldfield.^[11] machine and the tests were performed in conformance with

cast at three different cooling rates was produced by casting at

1440 °C in the form of ingots. They shall henceforth be desig-

1440 °C in the form of ingots. They shall henceforth be desig-

1440 °C in the form of ingot

and water-cooled chills (specimens A and B, respectively). (The tion $(5\times)$ using Steads reagent as the etchant.

with austenite. The minimum eutectic temperature can be used case no copper chill was used.) To make the mold for casting, as a measure of undercooling and reflects the amount of primary a teakwood pattern of size 225 by 150 by 25 mm was employed austenite precipitated and the chilling tendency of the cast iron. with standard pattern allowances. The molds were prepared Austenite dendrite interaction has been shown to be a major using silica sand with 5% bentonite as a binder and 5% moisture. factor in determining the UTS of irons.^[8] The molten alloys were cast in the mold and were cooled from The most important fact to recognize regarding cell structure one end by the chill. In the case of water-cooled chilling and is that it develops after the precipitation of austenite dendrites. liquid-nitrogen-cooled chilling, arrangements were made in the Thus, cell growth conforms to and is superimposed on a den-
copper chill to circulate water (at 23 °C) and liquid nitrogen

American Foundryman's Society standards. Each result of the test recorded in this paper was an average of at least three **2. Experimental Procedure** Fracture Fracture Experimental Procedure Fracture toughness tests were carried out on ASTM standard

2.1 Fabrication of Material 2.1 *Material Calcerimens Calcerimens* (with a chevron notch in the middle), all of which **2.1** *Fabrication of Materials properties properties properties properti* A cast iron alloy of the composition shown in Table 1 and Testing System. All tests were done in accordance with ASTM

the different cooling rates was produced by casting at 399-1990 test standards. A three-point bend frac

2.2 Casting Procedure 2.2 Casting Procedure Case of Crain size measurements were done on polished specimens Figure 1 shows a schematic diagram of the mold used for using ASTM index methods, and eutectic cell count measureproducing the various ingots cast with liquid-nitrogen-cooled ments were carried out on polished specimens at low magnifica-

Specimen location	Pearlite content (%)			Specimen location		UTS (MPa)	
Distance from chill end (mm)	Specimen A	Specimen в	Specimen	Distance from chill end (mm)	Specimen A	Specimen B	Specimen
30	28	32	59	30	488.67	467.33	282.55
70	35	39	58	70	381.44	328.34	282.43
105	42	60	61	105	372.86	361.43	287.53
145	57	62	59	145	272.43	279.31	281.54
180	59	68	60	180	268.32	282.67	279.75
225	68	71	59	225	273.64	281.54	282.65

Table 2 Pearlite content of various specimens along the Table 5 UTS various specimens along the length of **length of the casting the casting the casting**

Table 3 Grain size of various specimens along the
 Table 6 Fracture toughness of various specimens along

the length of the casting

the length of the casting

		Number of grains/in. ² at $100 \times$		Specimen location	Fracture toughness (MPa \sqrt{m})		
Specimen location	magnification			Distance from chill	Specimen	Specimen	Speci
Distance from chill end (mm)	Specimen A	Specimen	Specimen	end (mm)	A		
				30	33.9	32.1	22.
30	128	128	32	70	31.2	28.4	21.
70	64	64	32	105	29.1	26.8	22.
105	64	32	32	145	28.7	24.1	20.
45	32	32	32	180	24.7	23.8	22.
180	32	32	32	225	23.6	23.0	22.
225	64	64	32				

and the distribution of the eutectic graphite. The nucleation and growth.

en location	Pearlite content (%)			Specimen location	UTS (MPa)		
e from chill (mm)	Specimen A	Specimen В	Specimen	Distance from chill end (mm)	Specimen	Specimen	Specimen
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Table 4 Eutectic cell count of various specimens along
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t compare the eutectic cell counts of water-cooled and liquid-
nitrogen-cooled cast iron with ordinary sand-cast iron. Pearlite content, UTS, and fracture toughness were plotted against eutectic cell count for various specimens along the length of the casting, as in Fig. 2 to 4, respectively. Some typical photomicrographs of eutectic cells for sand-cast, water-cooled, and liquidnitrogen-cooled chilled cast iron are shown in Fig. 5 to 10. It
can be seen from Fig. 3 and 4 that the UTS and fracture
toughness increase as the eutectic cell count increases until the cells become compacted (as shown in Fig. 5 to 10). It can be seen that the difference between the eutectic cell counts for water-cooled and liquid-nitrogen-cooled chilled cast iron is **3. Results and Discussion** small, whereas that between the eutectic cell counts for chilled cast iron and sand-cast iron is very large. Typical values are Table 2 to 6 show the pearlite content, the grain size, the energy and the fracture toughness, respectively, of the three specimens along the length of the casting.

The energy of the three specimens along the length of t

3.1 Eutectic Cell Count 3.1 Count Cell Count Eutectic Cell Count EUTEC EUTEC EUTEC EUTEC EUTEC Solidification of hypereutectic cast iron begins with the crys- cooled chilled cast iron. However, there are significantly fewer tallization of primary graphite. The primary graphite develops nuclei for solidification in the case of sand-cast iron without a as straight graphite plates with some branching growing while chill. This is due to the undercooling effect caused by chilling. totally surrounded by liquid. The composition of the remaining Perhaps the most important fact to recognize regarding cell liquid shifts toward the eutectic where the liquid is believed to structure is that it develops after the precipitation of austenite solidify in a manner similar to that of a eutectic cell, although dendrite structure, which solidifies first. The dendritic structure the primary graphite may influence the size of the eutectic cell hosts the cells and therefore can have a major effect on cell

Fig. 2 Plot of pearlite content vs eutectic cell count for various specimens along the length of casting

3.2 Rate of Cooling

The differences in associated graphite, its randomness,

length (as shown in Fig. 11 to 13), and matrix structure (as shown

shown in Fig. 11 to 13), and matrix structure (as shown

in Fig. 14 to 16), for the various speci interaction. Hence, the eutectic cells solidify around these aus-
tenite dendrites, and in this manner the entire microstructure **3.3 Pearlite Content** is affected by the number and size of the dendrites. Also, if the results shown in Table 2 to 4 are compared with

Fig. 3 Plot of UTS vs eutectic cell count for various specimens along the length of casting

Therefore, the general picture of solidification of chilled one another, it can be seen that there exists a straightforward

Fig. 4 Plot of fracture toughness vs eutectic cell count for various specimens along the length of casting

taken along the length of the casting for the various specimens cells together. Since dendrites are formed from primary austencast. This relationship is plotted in Fig. 2. Comparing Table 2 ite, they have a higher tensile strength than eutectic carbon. and 6, it can be deduced also that fracture toughness decreases Here, due to chilling, dendrites directly form austenite and do with an increase in pearlite content and *vice versa*. This too not contain flake carbon. Instead they contain carbides (cementagrees with Baker's findings.^[14] It is noteworthy that specimens ite) in the pearlite matrix, as shown in Fig. 15 and 16. Baker^[4] A and B contain cementite in pearlite matrix, whereas specimen in his investigation on fracture mechanics of gray cast iron C contains ferrite in pearlite matrix (Fig. 14 to 16). observed that both tensile strength and fracture toughness

Fig. 6 Eutectic cell structure of water-cooled chilled specimen 30 mm from the chill end, etched with Steads reagent (165 cells/cm² at $5\times$ magnification)

Fig. 7 Eutectic cell structure of water-cooled chilled specimen 225 mm from the chill end, etched with Steads reagent (70 cells/cm/² at $5\times$ magnification)

3.4 Dendrite Formation

Results of the investigation indicate that the minimum eutectic temperature can be used as a measure of undercooling and **Fig. 5** Eutectic cell structure of sand-cast specimen, etched with reflects the number of cells precipitated and the chilling ten-
Steads reagent (75 cells/cm² at 5× magnification) dency of the iron. The effects of eute dency of the iron. The effects of eutectic cell count on UTS and fracture toughness of the specimens tested in this investigation are shown in Fig. 3 and 4. Higher dendritic interaction areas in the case of chilled cast irons reflect the interweaving relationship between pearlite content and eutectic cell count of dendrite through eutectic cells that effectively tie the eutectic

men 30 mm from the chill end, etched with Steads reagent (185 cells/ men 225 mm from the chil
cm² at 5× magnification) cm² at 5× magnification cm² at $5\times$ magnification)

Fig. 8 Eutectic cell structure of liquid-nitrogen-cooled chilled speci-
men 30 mm from the chill end, etched with Steads reagent (185 cells/ men 225 mm from the chill end, etched with Steads reagent (70 cells/

Fig. 9 Eutectic cell structure of liquid-nitrogen-cooled chilled speci-
end, unetched (type A graphite, 250 \times magnification) men 105 mm from the chill end, etched with Steads reagent (105 cells/ cm² at $5\times$ magnification)

decreased linearly with increasing eutectic cell size (smaller number of cells/cm²), and this agrees well with the results of **3.6 Mode of Fracture** the present investigation, which indicate that UTS and fracture
toughness increase as the cell size decreases (larger number of the fracture surfaces of specimens revealed a brittle mode of
cells/cm²).

size. It can be seen that the grains in the case of chilled cast specimen C has the lowest UTS and fracture toughness, foliron were fine but the grains were very coarse in the case of lowed by specimens B and A in that order. This means that an sand-cast iron cast without a chill. The fine grain size in the increase in the rate of chilling results in an increase in UTS case of chilled cast iron results in the soundness of the casting and fracture toughness of the material. and hence its high strength and toughness. In water-cooled and Moreover, the further away form the chill the specimen is liquid-nitrogen-cooled chilled cast iron, the experimental data taken, the lower the UTS and fracture toughness.^[13] This is show that dendrite morphology is the principal factor in obviously because the further away form the chill the specimen

Fig. 11 Microstructure of sand-cast specimen 30 mm from the chill

determining the strength and toughness but not the graphite morphology, an observation supported by Read Hill.^[12]

). fracture near the chill end. Looking at Table 5 and 6, it can be **3.5 Grain Size** Size **3.5 Grain** Size mechanical properties are the rate of chilling and the speciment of chilling and the specim Table 3 shows the results of microscopic studies on grain location. It is observed that, if all other factors are kept constant,

Fig. 12 Microstructure of water-cooled chilled specimen 30 mm from 30 mm from the chill end, etched with Nital (cementite in pearlite, the chill end, unetched (type D graphite, 250× magnification) 500× magnification) the chill end, unetched (type D graphite, $250\times$ magnification)

Fig. 15 Microstructure of water-cooled chilled cast iron specimen

Fig. 13 Microstructure of liquid-nitrogen-cooled chilled cast iron **Fig. 16** Microstructure of liquid-nitrogen-cooled chilled cast iron

specimen 30 mm from the chill end, unetched (type D graphite, specimen 30 mm from the chill end, etched with Nital (massive cement-
250 \times magnification) its in pearlite plus some ledeburite, 500 \times magnification) ite in pearlite plus some ledeburite, $500\times$ magnification)

Fig. 14 Microstructure of sand-cast specimen 30 mm from the chill men 30 mm from the chill end, indicating cleavage fracture of graphend, etched with Nital (ferrite in pearlite, 500 × magnification) ite flakes end, etched with Nital (ferrite in pearlite, $500 \times$ magnification)

Fig. 17 SEM fractograph of sand-cast fracture toughness test speci-

Fig. 19 SEM fractograph of liquid-nitrogen-cooled fracture tough- 7. H. Fedriksson and M. Hillert: *Br. Foundryman*, 1971, Feb., p. 54. ness test specimen 30 mm from the chill end, indicating dimpled rupture 8. D. Gloveratal: *Trans. AFS*, 1982, vol. 90, p. 745. of graphite flakes 9. R.W. Heine and C.R. Loper: *Trans. AFS*, 1969, vol. 77, p. 185.

is, the lower is the rate of chilling. This agrees with the deduc-

tions made earlier that increasing the rate of undercooling due

13. K.W.H. Seah, J. Hemanth, and S.C. Sharma: *J. Mater. Sci.*, 1995, vol. to chilling tends to result in an increase in UTS and fracture 30, p. 4986.
toughness of the material. 14. M.O. Spiege

4. Conclusions

Experiments have been performed using the number of eutectic cells as an index of graphite nucleation and the effect of these on structure and properties, since these cells exist in molten cast iron to act as nuclei of graphite during solidification. The following conclusions were made.

- The number of eutectic cells is significantly larger in the case of chilled cast iron than in the case of sand-cast iron cast without a chill.
- It has been established that the number of eutectic cells affects both the structure and properties of the material. Fig. 18 SEM fractograph of water-cooled fracture toughness test
specimen 30 mm from the chill end, indicating dimpled rupture of
graphite flakes
graphite flakes
will be the UTS and fracture toughness. In the cast iron
will
	- Data on irons chilled using water-cooled and liquid-nitrogen-cooled chills indicated that the cooling rate has a marked effect on eutectic cell size, grain size, microstructure, and mechanical properties.

A straightforward relationship exists between pearlite content and the eutectic cell count.

References

- 1. R.T. Wimber: *Trans. AFS*, 1980, vol. 88, p. 717.
- 2. D.O. Morton: *Foundry Practice-Foseco*, 1979, Dec. (200), p. 138.
- 3. D.M. Stefunescu and C.R. Loper: *Giesseri*, 1981, Mar. (5), p. 73.
- 4. T.J. Baker: *Trans. AFS*, 1978, vol. 88, p. 13.
- 5. K. Suzuki and N. Kayama: *Trans. AFS*, 1982, vol. 90, p. 423.
- 6. R.T. DeHoff and F.N. Rhines: *Trans. AFS*, 1982, vol. 90, p. 423.
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- 10. G.F. Ruff and J.F. Wallace: *Trans. AFS*, 1977, vol. 85, p. 179.
- 11. J.Y. Dawson and W. Oldfield: *BCIRA J.*, 1960, vol. 8 (2), p. 221.
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- 14. M.O. Spiegeletal: *Ind. Foundry J.*, 1984, vol. 35, p. 15.